

# Reduction-Induced Synthesis of Reduced Graphene Oxide-Wrapped Cu<sub>2</sub>O/Cu Nanoparticles for Photodegradation of Methylene Blue

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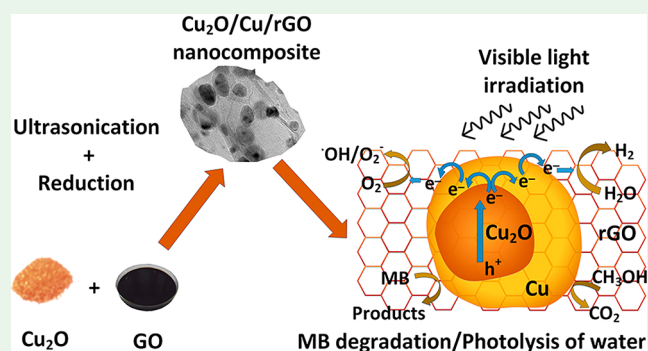
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**ABSTRACT:** A photocatalytic composite in powder form having nanosized Cu<sub>2</sub>O/Cu granules wrapped in multilayered reduced graphene oxide (rGO) sheets was synthesized. The fabrication process was initiated by potentiostatic electrodeposition of n-type Cu<sub>2</sub>O thin films on Ti substrates in an acetate bath. Peeled-off Cu<sub>2</sub>O thin films were next subjected to ultrasonic mixing with graphene oxide (GO) by maintaining a Cu<sub>2</sub>O/GO mass ratio of 125:10. Chemical reduction using ascorbic acid converted Cu<sub>2</sub>O partially to Cu, while converting GO completely to rGO as confirmed by Fourier transform infrared spectroscopy (FTIR), X-ray photoelectron spectroscopy (XPS), and X-ray diffraction (XRD). Transmission electron microscopy (TEM) images of the composite confirmed the presence of Cu nanoparticles of the size of tens of nanometers within the photocatalyst, Cu<sub>2</sub>O. The fabrication process therefore implicitly favors synthesis of nanosized Cu playing the role of a co-catalyst. The Cu<sub>2</sub>O/Cu/rGO composite showed significantly higher ability to degrade methylene blue in solution, while the effects due to photocorrosion were minimal during a month of testing when compared with Cu<sub>2</sub>O alone. The layered rGO provides an effective medium for transporting photoelectrons to the reactant while inhibiting photocorrosion. Furthermore, the composite demonstrated its ability to split water under visible light in the presence of methanol and creates a promising platform to further develop a variety of other photocatalytic applications.

**KEYWORDS:** photocatalyst, cuprous oxide, reduced graphene oxide, nanocomposite, co-catalyst



## 1. INTRODUCTION

Development of photocatalytic materials in the form of harnessing solar energy has been intensely researched given its potential to drastically reduce the anthropogenic carbon footprint.<sup>1</sup> Sustainable use of material, scalable synthesis, biosafety, and abundance of material have become integral and important considerations in the material development process. Metal oxide semiconductors acting as photoharvesters form a broad class of photocatalytic materials, and TiO<sub>2</sub>, ZnO, WO<sub>3</sub>, CuO, and Cu<sub>2</sub>O are some of the most cost-effective photocatalytic materials being researched.<sup>2–6</sup> Optimization of material properties enabling efficient production of electron–hole pairs from abundant solar photons, effective charge separation and transportation with minimal intermediate losses, resistance to photocorrosion, and long-term stability of the material, especially under aqueous environments, have become critically important, while careful choice of an accompanying metal co-catalyst can further enhance the associated photocatalytic redox reaction.<sup>7,8</sup> Fujishima and Honda in 1972 realized the first ever electrochemical photolysis of water using TiO<sub>2</sub>,<sup>9</sup> and over the years the

material has demonstrated its promise as an excellent photocatalyst in a variety of photocatalytic applications.<sup>10–16</sup> However, its wide band gap (>3.0 eV)<sup>11,13,16</sup> only allows harvesting the ultraviolet end of the solar spectrum,<sup>17</sup> making its photoelectron yield slightly low for terrestrial applications.

The placement of redox potential with respect to the semiconductor conduction and valence band (VB) energy levels has also become important in order to sustain a specific photocatalytic reaction. Therefore, band-gap re-engineering through doping, noble metal loading, or dye sensitization has become a practical requirement for enhancing the photocatalytic activity of a material.<sup>18–21</sup> However, such specialized treatment may seriously limit mass-scale production of the

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material given the high cost of production.<sup>22,23</sup> Alternatively, semiconductors with a narrower band gap, for example,  $\text{Cu}_2\text{O}$ , which is one of the best metal oxides suited for terrestrial photocatalytic applications, have become attractive given the materials' earth abundance, nontoxicity, and compatibility with a variety of cost-effective scalable fabrication techniques.<sup>2,6</sup> Similar photocatalytic metal oxides have since been reported and used in a variety of important photoelectrochemical reactions: for example, reduction of  $\text{CO}_2$  in the presence of water, degradation of organic pollutants in air or aqueous media, and removing heavy metals from water.<sup>13,24,25</sup> Synthesis of nanostructured photocatalytic materials has also attracted great attention due to the remarkable enhancement of physical and chemical properties when compared to their bulk counterpart. The increase in the surface-to-volume ratio by nanostructuring not only enables harvesting of more photons, resulting in more photoinduced electron–hole pairs, but also its nanoscale surface morphology offers shorter charge transport pathways and more redox active sites, leading to increased reaction kinetics. On the contrary, the larger surface area promotes increased defect sites at the grain boundaries, leading to increased charge recombination. Therefore, nanostructuring of materials with proper optimization remains critical to simultaneously maximize the harvested photoelectrons as well as improve the charge transport properties. To reduce undesirable electron–hole recombination or to gain high quantum efficiency, photocatalysts are often decorated with metal nanoparticle co-catalysts (Pt, Pd, Au, Rh, etc.).<sup>8</sup> These noble-metal nanoparticles improve charge separation and migration by forming a Schottky barrier between them and the semiconductor. It has also been reported that the metal nanoparticles such as Au and Ag exhibit strong surface plasmon resonance under visible light, significantly aiding photoharvesting of charge carriers.<sup>20,26,27</sup> Although somewhat susceptible to oxidation, due to earth abundance and low cost, non-noble metals such as Cu and Al have also attracted attention as co-catalysts.<sup>28</sup> In the present article, we report the preparation of  $\text{Cu}_2\text{O}$  thin films using electrodeposition, which enables the precise control of film morphology, crystallite size, electronic properties, and crystal facets. Furthermore, utilizing reduced graphene oxide (rGO) is explored to minimize photocorrosion.  $\text{Cu}_2\text{O}$  is naturally a p-type semiconductor due to the presence of defects such as copper vacancies or extra oxygen. However, electrodeposition brings flexibility to enable fabrication of both n-type and p-type  $\text{Cu}_2\text{O}$  films by controlling the deposition bath conditions.<sup>29</sup> Various methods, such as reduction of particle size and modification of surface structure, have been utilized in improving the photocatalytic activity and stability of  $\text{Cu}_2\text{O}$ -based photocatalysts.<sup>30</sup> However, the use of  $\text{Cu}_2\text{O}$  as a photocatalyst has been heavily limited due to its susceptibility to photocorrosion and carrier recombination.<sup>11</sup>

Graphene is a single layer of  $\text{sp}^2$ -hybridized carbon atoms in a hexagonal honeycomb lattice<sup>13,15,31–33</sup> and possesses characteristic features such as extremely high electron mobility, heat conduction, permeability to light, tensile strength, hydrophobicity, impermeability to gases and liquids, and ability to engineer surface functionalization.<sup>34</sup> Alternately, converting graphene oxide (GO) to reduced graphene oxide (rGO) simply removes oxygen moieties from GO, leading to properties close to those of graphene. Layering of rGO on the photocatalyst therefore has been considered as a viable option to improve stability against photocorrosion and has been

tested on  $\text{Cu}_2\text{O}$ .<sup>35</sup> Chemical reduction of GO is one of the easier and cheaper routes for producing rGO, and is used for layering of material in large-scale production and as an alternative to pristine graphene with a variety of options for specific chemical functionalization.<sup>34</sup> Several studies on the synthesis of  $\text{Cu}_2\text{O}/\text{rGO}$  composites using chemical reduction have shown to enhance photocatalytic efficiency compared to pure  $\text{Cu}_2\text{O}$ ; hence rGO has shown to serve as a useful electron acceptor material when combined with  $\text{Cu}_2\text{O}$ .<sup>14,36,37</sup> One of the earliest such work by Tran et al. reported the use of glucose for chemical reduction and the synthesized composite showed the ability to produce hydrogen.<sup>38</sup> Alternative chemical reduction routes have been deployed in the production of  $\text{Cu}_2\text{O}/\text{rGO}$  composites through the use of hydrazine hydrate,<sup>39,40</sup> wet reduction by using ascorbic acid,<sup>13</sup> and by using ethylene glycol.<sup>37</sup> Common to all of the above fabrication methodologies was the use of a solution of a copper salt and graphene oxide (GO) as starting materials. Out of the many options available for chemical reduction, only few methods are environmentally friendly.<sup>13</sup> Chemical reaction, photoreduction followed by solvothermal synthesis, and laser irradiation strategies are ubiquitous synthesizing methods for loading Cu nanoparticles on semiconductor photocatalysts.<sup>28</sup> Considering the above strategies, the method presented in this article can be mentioned as a simple way to decorate nanosized  $\text{Cu}_2\text{O}$  with Cu domains wrapped with rGO sheets without any external precursors. Once the peeled-off electrodeposited  $\text{Cu}_2\text{O}$  was added to GO, ultrasonication enabled both the homogenization as well as formation of nanosized  $\text{Cu}_2\text{O}$  particulates within the GO solution. Eventually,  $\text{Cu}_2\text{O}/\text{rGO}$  composites were synthesized through the reduction of GO to rGO and partial reduction of  $\text{Cu}_2\text{O}$  to Cu by using ascorbic acid. Consequently, the composite was tested for its photocatalytic ability to degrade methylene blue (MB) and to produce  $\text{H}_2$  through water splitting.

## 2. EXPERIMENTAL SECTION

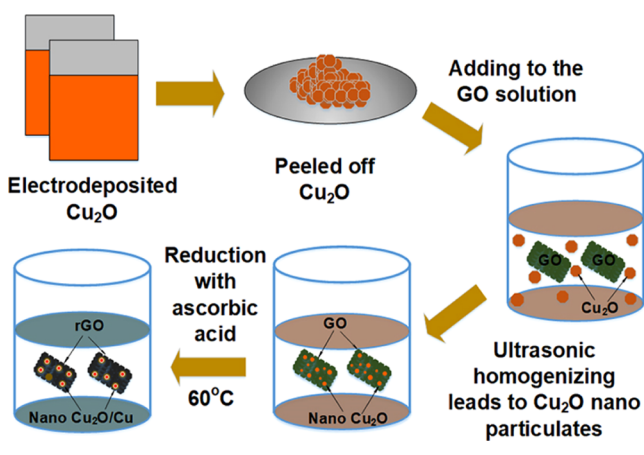
**2.1. Preparation of Graphene Oxide.** Graphene oxide (GO) was prepared from graphite powder (Sri Lanka Bogala vein graphite) by the improved Hummer's method.<sup>41</sup> In brief, a 9:1 mixture of concentrated  $\text{H}_2\text{SO}_4$  (Sigma-Aldrich, purity 95.0–99.0%)/ $\text{H}_3\text{PO}_4$  (Sigma-Aldrich, purity > 85.0%) and a mixture of graphite powder (18 g) and  $\text{KMnO}_4$  (Sigma-Aldrich, purity > 99.0%) (6 g) were prepared. Then, the mixture of graphite powder and  $\text{KMnO}_4$  was gradually added into the mixture of  $\text{H}_2\text{SO}_4/\text{H}_3\text{PO}_4$ . As this was an exothermic reaction, the temperature was maintained below 30 °C using an ice bath. Subsequently, the mixture was heated up to 50 °C and stirred for 12 h. Then, the mixture was allowed to cool to room temperature, and a mixture of ice (~400 mL) and 30%  $\text{H}_2\text{O}_2$  (10 mL) was used to terminate the oxidation reaction. The mixture was then kept for precipitation and washed several times until its pH was around 7. Finally, a 25 mL solution of GO was poured into a Petri dish and dried to obtain a GO membrane.

**2.2. Electrodeposition of Cuprous Oxide.**  $\text{Cu}_2\text{O}$  was electrodeposited on Ti substrates in an acetate bath containing 0.1 M sodium acetate (Sigma-Aldrich, purity > 99.0%) and 0.01 M cupric acetate (Sigma-Aldrich, purity > 99.0%) at a potential of –200 mV with respect to the reference electrode. Initially, Ti substrates were polished with sandpaper, cleaned with a detergent, and diluted with HCl and distilled water. Then, the substrates were ultrasonically cleaned in a distilled water bath. The electrodeposition was carried out using a three-electrode system and a CS 150 CorrTest Electrochemical Workstation. A platinum plate and a saturated calomel electrode were used as the counter electrode and the reference electrode, respectively. The pH of the deposition bath was measured using a SPER Scientific pH meter and adjusted using

sodium hydroxide and acetic acid in order to obtain  $\text{Cu}_2\text{O}$  with the desired grain size and crystallite facets.  $\text{Cu}_2\text{O}$  films with micro-sized polycrystalline grains were obtained at pH 6.2.<sup>29</sup> The  $\text{Cu}_2\text{O}$  films prepared by this process have n-type conductivity.<sup>42</sup> After the deposition, samples were kept at room temperature for drying and then peeled off gently to be collected as  $\text{Cu}_2\text{O}$  powder.

**2.3. Preparation of  $\text{Cu}_2\text{O}/\text{rGO}$  Composites.** Different amounts of GO (1, 2.5, 5, 10, 20, and 30 mg) were added into 10 mL of distilled water and ultrasonically dissolved using a TF-650Y ultrasonic homogenizer. Next, 125 mg of  $\text{Cu}_2\text{O}$  powder was added into each GO solution and kept in the ultrasonic homogenizer for 30 min to disperse well. Then, 10 mL of ascorbic acid (0.1 M) (Sigma-Aldrich, reagent grade, crystalline) solution was added into a mixer and stirred at 60 °C for different time intervals (10, 20, 30, 40, 50, and 60 min). The fabrication process is clearly indicated in Scheme 1. Finally, the formed precipitate was washed several times with water and ethanol. The product was then dried at 60 °C.

**Scheme 1. Illustration of the Formation Process of the  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  Nanocomposite**



**2.4. Characterization of  $\text{Cu}_2\text{O}/\text{rGO}$  Composites.** The surface morphology of the composite was investigated by scanning electron microscopy (SEM) (Zeiss Evols15). Inner structures of the composite were characterized by a transmission electron microscope (TEM, H-800). The samples for TEM were prepared by dispersing the product in ethanol using a sonicator bath for 15 min and dropped on a copper grid for observation. The crystal structure was confirmed by X-ray powder diffraction (XRD) (Rigaku Ultima IV). X-ray photoelectron spectroscopy (XPS) measurements were performed using an ESCALAB 250Xi spectrometer. All of the binding energies were calibrated to the reference of the C 1s peak at 284.6 eV. Fourier

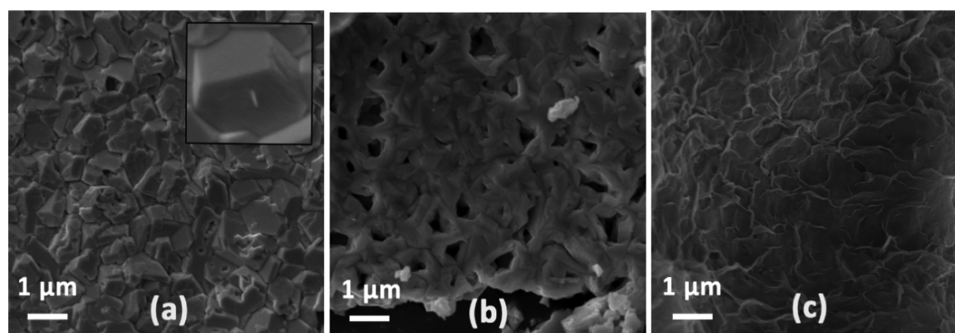
transform infrared (FTIR) spectra of the samples were obtained using a Thermo Scientific Nicolet iS10 spectrometer.

**2.5. Photocatalysis Measurements.** **2.5.1. Photodegradation of Methylene Blue (MB).** The photocatalytic activity of  $\text{Cu}_2\text{O}/\text{rGO}$  composites was investigated using the photodegradation of MB in water. The catalyst (20 mg) was mixed into 50 mL of MB (10 mg/L) solution and stirred well for 30 min in the dark. Then, the solution was irradiated using a 300 W xenon lamp (the photon flux at the reactor was 3.66  $\text{mW}/\text{cm}^2$ ). The reaction mixture was stirred continuously using a magnetic stirrer. At given intervals, 1 mL portions of the reaction mixture were drawn out for analysis. These samples were centrifuged to remove the dispersed catalyst powder and analyzed for MB by using ultraviolet–visible (UV–vis) spectroscopy (Ocean Optics, FLAME-T-UV–VIS–ES).

**2.5.2. Photocatalytic  $\text{H}_2$  Production.** The  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  composites were produced at selected reduction durations (20, 30, and 40 min) and evaluated for their ability to produce  $\text{H}_2$ . A mixture of 20 mL of methanol and 80 mL of water was used in the reactor with 20 mg of the composite. The reactor was maintained at 25 °C during the evolution of  $\text{H}_2$ . The gas products were analyzed and quantified using GC Smart (GC-2018) Shimadzu gas chromatography (GC). After every 60 min, 200  $\mu\text{L}$  of gas was taken from the samples using a gas-tight syringe and was injected into a gas chromatograph to identify the gaseous products during the photocatalytic reaction using a thermal conductivity detector with argon as the carrier gas. The sample was irradiated by a Xe lamp (300 W). The light intensity was measured with a thermophile sensor (818P-001-12, Newport) and fixed at 100  $\text{mW}/\text{cm}^2$  by controlling the distance between the reactor and the lamp.

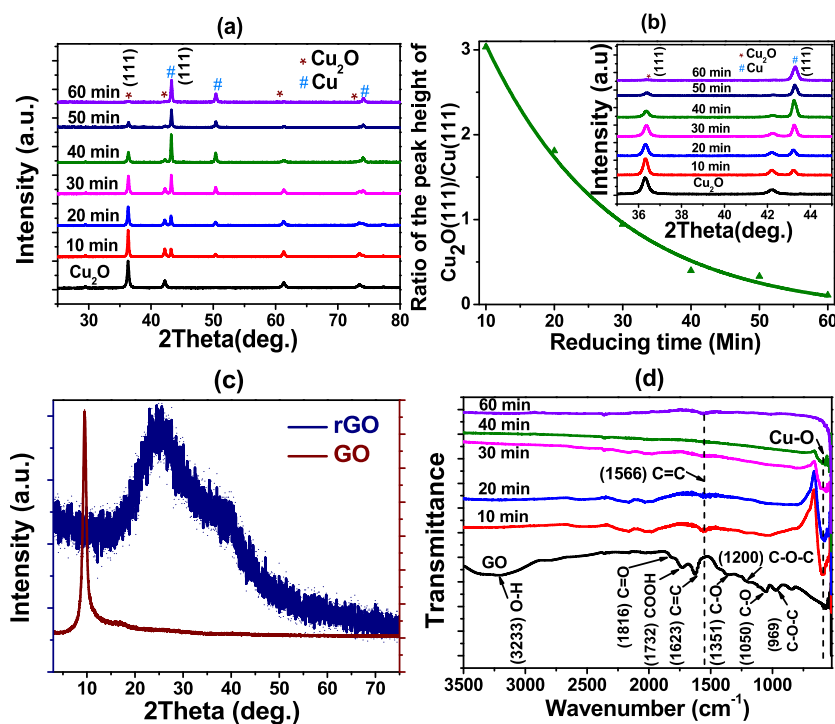
### 3. RESULTS AND DISCUSSION

Electrodeposition allows control of both the conductivity type and the surface morphology of the deposited films by controlling the pH of the deposition bath. A detailed study on the change of conductivity type of the electrodeposited  $\text{Cu}_2\text{O}$  with bath pH has been conducted.<sup>29</sup> Figure 1a shows an SEM image of an electrodeposited  $\text{Cu}_2\text{O}$  thin film deposited at pH 6.2. When electrodeposited at pH 6.2,  $\text{Cu}_2\text{O}$  has n-type conductivity and the alignment of bands referenced to vacuum are reported to be at  $-4.19$  and  $-6.29$  eV for the conduction band (CB) and the valence band, respectively.<sup>29</sup> The morphological features of the deposited thin films show micro-sized grains dominated by truncated octahedral shapes. It can be seen that the surface morphology of the peeled-off films has changed due to the release of strain, as shown in Figure 1b. However, it can also be seen that, on average, the grain sizes remain the same. The synthesized  $\text{Cu}_2\text{O}/\text{rGO}$  composite in



**Figure 1.** SEM images of  $\text{Cu}_2\text{O}$  thin films (a) electrodeposited on a Ti substrate in an acetate bath containing 0.1 M sodium acetate and 0.01 M cupric acetate at pH 6.2 for 40 min (the inset shows the zoomed image of the  $\text{Cu}_2\text{O}$  grain); (b) the same film after being peeled off from the substrate; and (c)  $\text{Cu}_2\text{O}/\text{rGO}$  composite in a powder form synthesized by ultrasonically homogenizing the peeled-off electrodeposited  $\text{Cu}_2\text{O}$  thin films and GO and subsequently reducing using 0.1 M ascorbic acid.





**Figure 2.** (a) XRD patterns comparing electrodeposited pure n-type  $\text{Cu}_2\text{O}$  with  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  composites synthesized by ultrasonically homogenizing peeled-off electrodeposited  $\text{Cu}_2\text{O}$  thin films and GO, subsequently being reduced using 0.1 M ascorbic acid for 10, 20, 30, 40, 50, and 60 min, respectively, and maintaining an initial GO/ $\text{Cu}_2\text{O}$  mass ratio of 10:125. (b) Variation of the XRD peak height ratio of  $\text{Cu}_2\text{O}(111)$  to  $\text{Cu}(111)$  in  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  composites with reduction time (the inset shows an enlarged image of the XRD spectral peaks related to the (111) phase of  $\text{Cu}_2\text{O}$  and Cu). (c) XRD patterns of as-synthesized GO using the improved Hummers' method and chemically synthesized rGO reduced by ascorbic acid for 30 min. (d) FTIR spectra of the synthesized GO and  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  composites produced at different reducing durations (10, 20, 30, 40, 50, 60 min) with ascorbic acid.

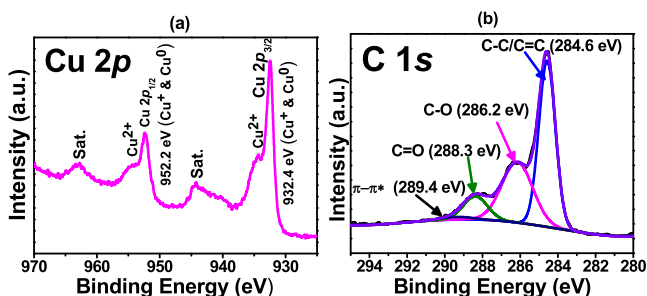
Figure 1c shows a more uniform surface morphology due to the formation of the rGO layer.

The structural properties of the composite evaluated by XRD measurements (see Figure 2a) showed the presence of  $\text{Cu}_2\text{O}$  from the observed characteristic peaks at  $29.4^\circ$  (110),  $36.3^\circ$  (111),  $42.18^\circ$  (200),  $61.26^\circ$  (220), and  $73.48^\circ$  (311), where the (111) peak was the most intense XRD peak. Furthermore, the composite prepared by reducing the  $\text{Cu}_2\text{O}$  and GO mixture retained the crystalline phase of  $\text{Cu}_2\text{O}$ , while showing the emergence of a  $\text{Cu}(111)$  peak. It can be observed that, with increasing reduction time, the intensity of the (111) XRD peak of  $\text{Cu}_2\text{O}$  decreases and that of the  $\text{Cu}(111)$  peak increases, confirming the reduction of  $\text{Cu}_2\text{O}$  to Cu with ascorbic acid. This indicates that the reaction time with ascorbic acid controls the ratio of  $\text{Cu}_2\text{O}/\text{Cu}$  phases in the composite. In order to capture the dynamics of the reduction process, the XRD spectral peak intensity ratio of  $\text{Cu}_2\text{O}(111)/\text{Cu}(111)$  vs reaction time is plotted (see Figure 2b). An exponential decay of the ratio  $\text{Cu}_2\text{O}(111):\text{Cu}(111)$  can be observed, while the  $\text{Cu}_2\text{O}(111)$  peak almost entirely disappears after reacting with ascorbic acid for 60 min. It was confirmed in a separate experiment using XRD that GO reduces to rGO when reacted with ascorbic acid in a duration less than 30 min. It could be observed that GO having a single XRD peak at  $9.5^\circ$  disappeared with the growth of a broader XRD hump around  $24.5^\circ$  due to the formation of rGO. The broadened peak further indicated that graphene had exfoliated into a few-layered sheet-like structure of rGO (see Figure 2c).

To further analyze the active functional groups of synthesized rGO, FTIR spectra were collected for  $\text{Cu}_2\text{O}/$

$\text{Cu}/\text{rGO}$  composites synthesized under varied reduction levels (see Figure 2d). It can be observed that the oxygen functionalities in the composite are initially apparent but subside with increased reduction. The FTIR spectrum of the synthesized GO showed a strong band around  $3233 \text{ cm}^{-1}$ , primarily associated with the stretching vibrations of the O–H group. Other features were assigned to the C=O carboxylic acid stretching at  $1732 \text{ cm}^{-1}$ , the C–OH stretching at  $1351 \text{ cm}^{-1}$ , the  $\text{sp}^2$  structure of the unoxidized graphitic domains at  $1623 \text{ cm}^{-1}$ , the C–O stretching of Ar–OH groups at  $1200 \text{ cm}^{-1}$ , and the C–O stretching at  $1050 \text{ cm}^{-1}$ . Reduction of the composite leads to a decrease in the absorption bands of the O–H group, C=O group, C–O group, and C–OH group. The C=C peak for GO typically occurs around  $1623 \text{ cm}^{-1}$ . However, the reduction of GO shifts the C=C related peak of rGO to  $1566 \text{ cm}^{-1}$ , which confirms the restoration of the  $\text{sp}^2$  carbon network, and the observations were consistent with that reported in the literature.<sup>43</sup> The absorption band at  $582 \text{ cm}^{-1}$  in the  $\text{Cu}_2\text{O}/\text{rGO}$  composite can be ascribed to the stretching vibration of Cu–O. This indicates the existence of  $\text{Cu}_2\text{O}$  at intermediate stages of the reduction process. When the reducing time reaches 60 min, the absorption peak of Cu–O disappears entirely, indicating the complete reduction of  $\text{Cu}_2\text{O}$  to Cu.

X-ray photoelectron spectroscopy (XPS) measurements were carried out to investigate the composition and chemical states of the  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  composite reduced using ascorbic acid for 30 min. The corresponding XPS spectra for Cu and C are shown in Figure 3. The spectral peaks positioned at 932.4 and 934.4 eV correspond to the  $\text{Cu } 2\text{p}_{3/2}$ , indicating the

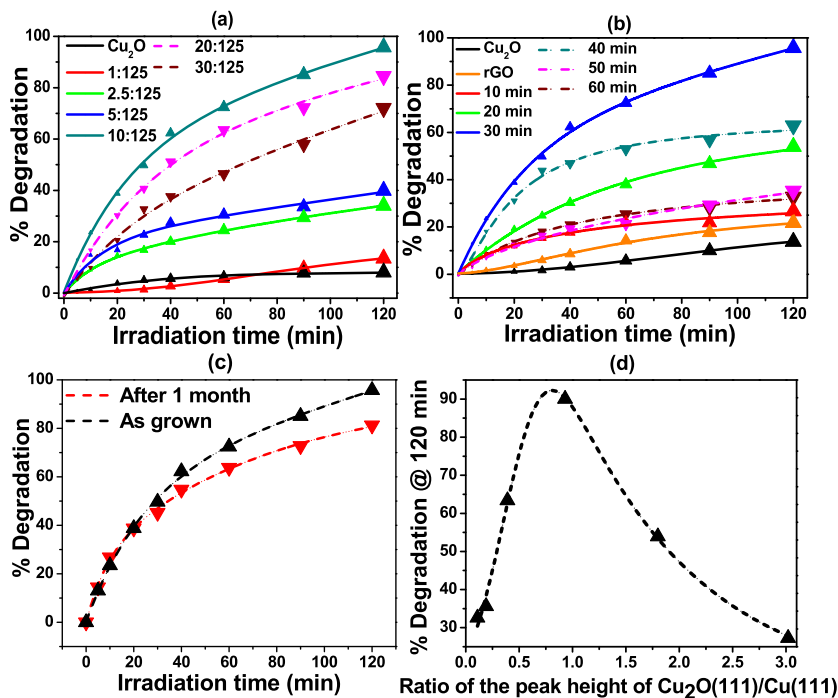


**Figure 3.** XPS spectra of (a) Cu 2p and (b) C 1s of the  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  composites synthesized by ultrasonically homogenizing peeled-off electrodeposited  $\text{Cu}_2\text{O}$  films and GO, subsequently being reduced using 0.1 M ascorbic acid for 30 min.

presence of  $\text{Cu}^+/\text{Cu}^0$  with a peak intensity of 38.7% and  $\text{Cu}^{2+}$  states with a peak intensity of 48.5%, respectively, in the composite (the binding energy resolution was insufficient to deconvolute the  $2p_{3/2}$  peaks of  $\text{Cu}^+$  and  $\text{Cu}^0$ ). The 952.2 eV ( $\text{Cu}^+/\text{Cu}^0$ ) and 954.4 eV ( $\text{Cu}^{2+}$ ) peaks (see Figure 3a) correspond to the Cu  $2p_{1/2}$ . Also, two satellite peaks appearing at 943.9 and 962.9 eV positions support the presence of the  $\text{Cu}^{2+}$  state. The presence of the  $\text{Cu}^{2+}$  state only in the XPS measurements indicates that the  $\text{Cu}^{2+}$  state is limited only to the surface of the formed composite particles. The absence of  $\text{CuO}$  or  $\text{Cu}(\text{OH})_2$  peaks in the XRD patterns clearly indicates that the bulk composite has not been influenced by the  $\text{Cu}^{2+}$  state. The obtained C 1s spectra were deconvoluted into four distinct peaks positioned at 284.6, 286.2, 288.3, and 289.4 eV having relative peak intensities of 52.7, 34, 10, and 3.2%, respectively. It was evident that the peak intensities at 289.4 eV

(indicative of  $\text{C}=\text{O}$ ) and 286.2 eV (indicative of  $\text{C}-\text{O}$ ) were both significantly lower compared to that of the peak at 284.6 eV (indicative of both  $\text{C}=\text{C}$  and  $\text{C}-\text{C}$ ). A similar comparison was made using the C 1s spectra reported for  $\text{GO}^{44}$  synthesized using Bogala vein graphite, the same precursor and the process used in our experiment. Clearly, the  $\text{C}=\text{O}$  and  $\text{C}-\text{O}$  associated peaks for GO were significantly more pronounced compared to the peak associated with  $\text{C}=\text{O}$  and  $\text{C}-\text{O}$ . This observation provided further evidence of reduction of GO in the synthesized composite and is consistent with the observations made on related work that used ascorbic acid as a reductant.<sup>45</sup>

The photocatalytic activity of the composite was tested by quantification of MB degradation. A xenon lamp was used to generate a broad-wavelength excitation source. A 10 mg/L MB solution was illuminated with the xenon source continuously for 2 h. After each predefined duration, small quantities of MB were extracted from the reactor and centrifuged to remove any residual photocatalyst. Thereafter, the absorption spectra were measured using a UV-vis spectrophotometer to evaluate the photodegradation of MB. The peaks of the collected absorption spectra were consistently located around 668 nm, which can be associated with the  $\pi$ -system of MB. However, to quantify the degradation of MB, the area subtended by the respective absorbance spectra in the range from 550 to 750 nm was measured in order to acquire a cumulative absorbance measure. Each of such estimates was then normalized with respect to the cumulative absorbance measured for the initial MB solution without illumination. The yielded normalized cumulative absorbance,  $A_{\text{abs}}$ , was thereafter converted to a percentage degradation fraction  $100(1 - A_{\text{abs}})$ . This

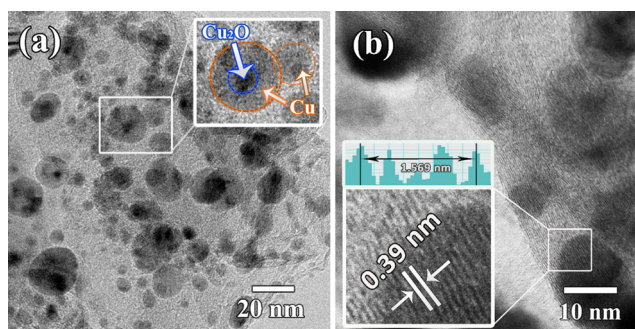


**Figure 4.** Percentage degradation of MB under the irradiation of visible light using a 300 W xenon lamp with  $\text{Cu}_2\text{O}/\text{rGO}$  composites (a) synthesized with different mass ratios of GO to  $\text{Cu}_2\text{O}$  (1, 2.5, 5, 10, 20, 30:125) and reduced using ascorbic acid (0.1 M) for 30 min, and (b) reduced with varied durations (10, 20, 30, 40, 50, 60 min) with 0.1 M ascorbic acid having an initial GO to  $\text{Cu}_2\text{O}$  mass ratio of 10:125. (c) Percentage degradation of MB under the irradiation of visible light with as-synthesized  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  and composite kept for 1 month. (d) Variation of the percentage degradation of MB at the irradiation time of 120 min with the ratio of the corresponding peak heights of  $\text{Cu}_2\text{O}(111)$  to  $\text{Cu}(111)$  extracted from the XRD patterns in Figure 2b. Solid lines show the best fits to the corresponding data points.

percentage degradation fraction was measured at predefined time stamps of 5, 10, 20, 30, 40, 60, 90, 120 min during illumination for each of the  $\text{Cu}_2\text{O}/\text{rGO}$  composites having different initial mass ratios of GO to  $\text{Cu}_2\text{O}$  (1, 2.5, 5, 10, 20, 30:125). Figure 4a confirms that increasing GO eventually leads to a decrease in the degradation efficiency, which can be attributed to the enhanced absorption and scattering of photons by the excess carbon present in the composite.<sup>46</sup> The highest degradation percentage was reported for the composite synthesized with a GO to  $\text{Cu}_2\text{O}$  ratio of 10:125 (see Figure 4a); hence, for subsequent synthesis of the composite and measurements the same ratio was maintained.

In order to further optimize the photocatalytic activity of the composite, MB degradation percentages were measured for the composites prepared using different reduction times (10, 20, 30, 40, 50, and 60 min). The photocatalytic activities of these composites were compared with the photocatalytic activities of  $\text{Cu}_2\text{O}$  and rGO separately. The composites showed significantly higher degradation ability of MB as compared to  $\text{Cu}_2\text{O}$  and rGO taken alone (see Figure 4b). The absorption spectra of MB degraded for different times using the composite (composite having the initial mass ratio of GO/ $\text{Cu}_2\text{O}$  10:125 and reduced for 30 min) are shown in the Supporting Information Figure S1. The inset of the figure shows the digital photographs of the MB solutions at different irradiation time intervals. Figure 4c compares the percentage degradation of MB (at different photocatalytic reaction times) of a freshly prepared composite with the same sample re-used after 1 month. Before being reused and tested, the sedimented composite was dried and stored for a duration of 1 month under ambient conditions. The percentage of degradation after one month was well within 20% of the initial value, indicating that the composite is relatively stable. It can be inferred that the layering of rGO would have significantly contributed towards the stability of the material. The claim can be further supported by some of the previous studies that investigated the decreased oxidation of  $\text{Cu}_2\text{O}$  under aqueous conditions after layering with rGO.<sup>38</sup> A reduction time of 30 min was found to produce the most active composite catalyst for the degradation of MB and this composite had a ratio of  $\text{Cu}_2\text{O}/\text{Cu} \approx 1:1$  as observed in the XRD patterns (Figure 4d).

To further probe the internal morphology of the material, TEM imaging was performed on  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  composites (synthesized with 30 min reduction using 0.1 M ascorbic acid and having an initial mass ratio of GO/ $\text{Cu}_2\text{O}$ =10:125) (see Figure 5). TEM images confirmed that the size distribution of the particles on average was around  $\sim 10$  to 20 nm, while thin sheets of rGO wrapped around the nanosized  $\text{Cu}_2\text{O}/\text{Cu}$  particles. The formation of ultrasmall particulates (with dimensions as small as 10 nm) was due to the rigorous sonication process introduced during the mixing of GO and  $\text{Cu}_2\text{O}$  prior to reduction. Starting from  $\text{Cu}_2\text{O}$  nanocolloids, synthesis of Cu nanoparticles using ascorbic acid-aided reduction has been previously reported.<sup>47</sup> The  $\text{Cu}_2\text{O}$  (in powder form) precursor used in our study was of much larger size (more than the microscale); therefore, it can be inferred that the ultrasonication is the predominant factor aiding the formation of nanosized Cu domains in the composite as observed by TEM. The TEM images in Figure 5a clearly show two distinct contrasting areas in the nanoparticles. These contrasting areas in TEM images can therefore be used to distinguish between Cu and  $\text{Cu}_2\text{O}$ .<sup>48</sup> The electron-dense Cu should lead to brighter regions, while the darker regions would

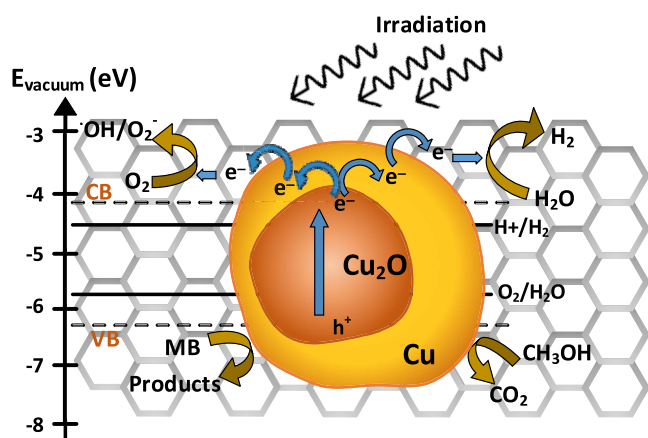


**Figure 5.** (a) TEM images of the  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  composite synthesized by ultrasonically homogenizing GO and peeled-off electrodeposited  $\text{Cu}_2\text{O}$  thin films (deposited on Ti substrates in an acetate bath of pH 6.2) having an initial mass ratio of GO/ $\text{Cu}_2\text{O}$ =10:125 and subsequently reduced using 0.1 M ascorbic acid for 30 min; the inset shows a typical  $\text{Cu}_2\text{O}/\text{Cu}$  nanoparticle; (b) an area of the composite showing many-layered rGO; the inset shows the interlayer separation of rGO sheets. Here, the average interlayer spacing was measured to be  $0.39 \pm 0.06$  nm.

correspond to  $\text{Cu}_2\text{O}$ . Most of the grains shown in the TEM images are indicative of the presence of nanosized Cu regions along with  $\text{Cu}_2\text{O}$  after the reduction process. Furthermore, several nanoparticles show only brighter areas indicative of some  $\text{Cu}_2\text{O}$  nanoparticles being completely converted into Cu during the reduction process. Previous studies show that these nanosized Cu domains could facilitate optical phenomena such as localized surface plasmon resonance (LSPR). Therefore, the effect could significantly assist photogenerated electrons to efficiently participate in the related redox reaction. Literature studies also show that Cu nanoparticles can produce LSPR in a long range of visible wavelengths, which can be tuned by controlling the particle size. From an electronic band perspective, the metal nanoparticles assist in excitation of electron–hole pairs between the d-band of the metal and the conduction band of the semiconductor (interband excitation).<sup>49–51</sup>  $\text{Cu}_2\text{O}/\text{Cu}$  nanoparticles densely scattered on the wrinkled graphene sheets are clearly observed in Figure 5a. A typical cross-sectional profile of a multilayered rGO sheet can be seen in Figure 5b, while further analysis yielded an inter-sheet separation of  $0.39 \pm 0.06$  nm consistent with the literature.<sup>52</sup> The composite also remained stable under aqueous conditions, and even after 1 month the material retained 80% of its ability to degrade MB, implying that the photocorrosion is minimal (see Figure 4c).

To summarize, an energy-level diagram showing the position of the bands of the electrodeposited  $\text{Cu}_2\text{O}$  is presented in Figure 6 with respect to the vacuum level found in our previous investigations.<sup>29</sup> Furthermore, related reactions for MB degradation and  $\text{H}_2$  generation by water splitting in the presence of methanol are depicted. When exposed to light,  $\text{Cu}_2\text{O}$  generates electron and hole pairs in the conduction band (CB) and valence band (VB). Reduction-induced Cu sites on the  $\text{Cu}_2\text{O}$  nanoparticles act as co-catalysts and improve the charge separation mechanism. These Cu nanoparticles also open a pathway to transfer the photogenerated electrons into rGO sheets with minimal losses. Also, the rGO sheets act as an electron acceptor and transporter, providing a path for the electrons to generate reactive oxidation species such as  $\cdot\text{OH}$  and  $\text{O}_2^-$ . Therefore, Cu nanoparticles facilitate minimization of the charge carrier recombination and help produce more reactive species. These reactive species promote the MB





**Figure 6.** Energy-level diagram and the electron transport in the degradation of MB and the photolysis of water in sacrificial mode with methanol, using  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  composite as a photocatalyst.

degradation into intermediate products. In order to evaluate the co-catalytic contribution of the nanosized Cu domains, a control experiment was performed. The percentage degradation of MB after 2 h of irradiation for nanosized  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  was as much as 70% higher compared to the composite without Cu (see Supporting Information Figure S2). Therefore, the critical role played by the metal co-catalyst was further highlighted by this fact.

When considering the  $\text{H}_2$  evolution reaction, the sacrificial donor, methanol, consumes holes in the VB by leaving an electron in the CB of  $\text{Cu}_2\text{O}$ . Partial reduction of  $\text{Cu}_2\text{O}$  creates Cu islands on  $\text{Cu}_2\text{O}$  particles, facilitating the separation of carriers and thus enhancing the photocatalytic activity. Graphene, having a high electron mobility due to its two-dimensional  $\pi$ -conjugated structure, acts as an electron collector and a transporter. It also separates the electrons and transfers them to reaction sites and improves  $\text{H}_2$  production. Graphene layers wrapped around the composite significantly limit self-reduction/oxidation and preserve  $\text{Cu}_2\text{O}$ , which is prone to photocorrosion. The composite was subsequently evaluated for its ability to split water into  $\text{H}_2$  and  $\text{O}_2$ . The  $\text{H}_2$  evolution reaction had to be facilitated using methanol as a hole scavenger. To measure the ability to evolve  $\text{H}_2$ , 20 mg of each composite reduced for 20, 30, and 40 min, respectively, was separately mixed in 100 mL of water/methanol mixture (volume ratio = 4:1). The evolved  $\text{H}_2$  was measured every hour using a gas chromatograph for a duration of 7 h (see Supporting Information Figure S3). The composite, reduced for 30 min, evolved  $0.9 \mu\text{mol}$  of  $\text{H}_2$  during 7 h, which was the highest. No significant changes were observed in the measured XRD spectra of the composite before and after 8 h of  $\text{H}_2$  evolution (see Supporting Information Figure S4), indicating that the structural properties of the composite remained stable during photocatalysis. The  $\text{H}_2$  evolution measurements were done under isothermal conditions in which the bath temperature was maintained around  $25^\circ\text{C}$ . Therefore, it should be noted here that the possibility of a methanol-reforming reaction, which occurs at much higher temperatures,<sup>53</sup> can be safely ruled out. Dye sensitization also plays an important role in the enhancement of photocatalysis.<sup>54</sup> During the MB degradation, this effect may further enhance the photocatalytic degradation. However, the composite's ability to evolve  $\text{H}_2$  by splitting water proves that any possible photocatalytic enhancement due to dye

sensitization is only supplementary and hence not a significant factor that determines the overall photocatalytic behavior.

#### 4. CONCLUSIONS

A nanosized  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  photocatalytic composite, sensitive to visible light, was synthesized using an electro-deposition-assisted chemical reduction route. Compared to pure, n-type  $\text{Cu}_2\text{O}$  thin films, the composite showed significantly high degradation ability of methylene blue in an aqueous medium. The composite was also stable under the tested conditions. As synthesis precursors, n-type  $\text{Cu}_2\text{O}$  thin films were initially electrodeposited on a Ti substrate and were peeled off before mixing with GO and distilled water. Ultrasonic homogenization converted the peeled-off n-type  $\text{Cu}_2\text{O}$  to nanosized particles dispersed in the GO solution. Consequently, the mixture was chemically reduced using ascorbic acid to produce the desired  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  photocatalytic composite. TEM images confirmed the presence of nanosized  $\text{Cu}_2\text{O}/\text{Cu}$  wrapped in multilayered rGO. Chemical reduction helped simultaneously to convert GO to rGO, while  $\text{Cu}_2\text{O}$  was partially converted to Cu. Best photocatalytic activity was reported for the composites prepared with an initial  $\text{Cu}_2\text{O}/\text{GO}$  mixture having a mass ratio of 125:10 and reduced using 0.1 M ascorbic acid for 30 min. The photocatalytic enhancement due to Cu in the composite was confirmed by comparing the percentage of MB degradation with that of the pure photocatalyst  $\text{Cu}_2\text{O}$  loaded on rGO. This enhancement can be attributed to the nanosized Cu acting as a co-catalyst and providing centers for efficient charge transfer. Furthermore, rGO in the composite acts as an electron acceptor with a smaller charge transfer resistance improving the interfacial electron transfer process. Therefore, both rGO and Cu contribute to the overall improvement of the photocatalytic activity of the composite. It is also worth noting that in this study, ascorbic acid, which is typically used for reducing GO, had been used simultaneously to reduce  $\text{Cu}_2\text{O}$  to Cu. This, as we perceive, is a novel route to load a co-catalyst in a photocatalyst with the least intervention, while making the fabrication process simpler. Though not comprehensively tested, it was experimentally confirmed that the composite produced  $\text{H}_2$  by splitting water under solar irradiation when the redox reaction was assisted by a sacrificial agent such as methanol. The composite therefore becomes an ideal platform to test many photocatalytic applications that heavily rely upon visible light.

#### ■ ASSOCIATED CONTENT

##### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsnm.0c03320>.

Absorption spectra of the  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  nanocomposite showing the time dependence of the amount of degradation in MB and the digital photographs of consequent discoloration of MB; comparison tests for the degradation of MB in the presence of the  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  nanocomposite,  $\text{Cu}_2\text{O}$ , rGO, and  $\text{Cu}_2\text{O}/\text{rGO}$ ;  $\text{H}_2$  evolution measurements of the  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  nanocomposites synthesized at different reduction durations (20, 30, 40 min); and XRD patterns of  $\text{Cu}_2\text{O}/\text{Cu}/\text{rGO}$  nanocomposite before and after the  $\text{H}_2$  evolution process (PDF)

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## Notes

The authors declare no competing financial interest.

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